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# High-pressure structural behavior of  $GdAIO<sub>3</sub>$  and  $GdFeO<sub>3</sub>$  perovskites

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#### Abstract

The orthorhombic perovskites, GdAlO<sub>3</sub> and GdFeO<sub>3</sub>, have been studied using single-crystal X-ray diffraction up to 8.52 and 8.13 GPa, respectively, in a diamond anvil cell at  $298 K$ . The evolution of the structures of GdAlO<sub>3</sub> and GdFeO<sub>3</sub> involves compression of both the GdO<sub>12</sub> and the octahedral (AlO<sub>6</sub> and FeO<sub>6</sub>) sites. The compression of the GdO<sub>12</sub> site is anisotropic in both perovskites, with the four longest Gd–O distances compressing more than the eight shorter Gd–O bond lengths, resulting in a decrease in the distortion of  $GdO_{12}$  with pressure. In  $GdAIO_3$ , the  $GdO_{12}$  site is less compressible than the  $AIO_6$  site, resulting in an increase of both the interoctahedral Al–O1–Al and Al–O2–Al angles with increasing pressure. Thus GdAlO<sub>3</sub> perovskite becomes less distorted with increasing pressure. In GdFeO<sub>3</sub>, the GdO<sub>12</sub> site displays a similar compressibility as the FeO<sub>6</sub> site, with little change in the Fe–O2–Fe angle with pressure but an increase of the Fe–O1–Fe tilting angle. Thus  $GdFeO<sub>3</sub>$  perovskite becomes less distorted with increasing pressure, but the change is not as pronounced as  $GdAIO<sub>3</sub>$ . The high-pressure behavior of  $GdAIO<sub>3</sub>$  and  $GdFeO<sub>3</sub>$  is similar to orthorhombic YAlO<sub>3</sub> perovskite but contrasts with orthorhombic CaSnO<sub>3</sub>, which becomes more distorted with increasing pressure.

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Keywords: GdAlO<sub>3</sub>; GdFeO<sub>3</sub>; Perovskite; Structures; High-pressure; X-ray diffraction

## 1. Introduction

The orthorhombic  $GdFeO<sub>3</sub>$ -type perovskites (*Pbnm*), with general stoichiometry  $ABO<sub>3</sub>$ , are derived from the ideal cubic structure  $(Pm3m)$  via the titling and distortion of the  $BO_6$  octahedra, as shown in [Fig. 1](#page-1-0) (e.g. Refs. [\[1–3\]\)](#page-7-0). The perovskites are of great interest in materials science because the relatively simple crystal structure displays many diverse electric, magnetic, piezoelectric, optical, catalytic, and magnetoresistive properties. In addition, perovskites are of interest in earth science because  $(Mg, Fe)SiO<sub>3</sub>$  transforms to a perovskite structure with Pbnm symmetry at high pressures and temperatures and is believed to form the bulk of the Earth's lower mantle (e.g. Ref. [\[4\]](#page-7-0)). Studies

\*Corresponding author. E-mail address: nross@vt.edu (N.L. Ross). of  $GdFeO<sub>3</sub>$ -type perovskites at high pressure therefore provide a useful probe to understand atomistic controls of structural changes.

Both GdFeO<sub>3</sub> and GdAlO<sub>3</sub> perovskite are Phnm perovskites, but with differing degrees of distortion from the ideal cubic structure. The greater distortion of  $GdFeO<sub>3</sub>$  relative to  $GdAlO<sub>3</sub>$  is reflected in the of GareO<sub>3</sub> relative to GaAlO<sub>3</sub> is reflected in the observed tolerance factor,  $t_{\text{obs}} = \langle A - O \rangle / \sqrt{2} \langle B - O \rangle$ , where  $\langle A-O \rangle$  and  $\langle B-O \rangle$  are the mean interatomic separations between twelve and six nearest neighbors for the  $A$  and  $B$  sites, respectively [\[5\]](#page-7-0). For a cubic perovskite,  $t_{obs} = 1$  but GdAlO<sub>3</sub> has  $t_{obs} = 0.986$  [\[6\],](#page-7-0) similar to  $CaTiO<sub>3</sub>$  perovskite [\[7\]](#page-7-0), and  $GdFeO<sub>3</sub>$  has  $t_{\text{obs}}=0.977$  [\[6\]](#page-7-0), similar to MgSiO<sub>3</sub> perovskite. The equations of state of  $GdAIO<sub>3</sub>$  and  $GdFeO<sub>3</sub>$  perovskite have been determined at room temperature (298 K) using high-pressure single-crystal X-ray diffraction [\[6\].](#page-7-0) The reported third-order Birch–Murnaghan equation of

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<span id="page-1-0"></span>

Fig. 1. Polyhedral representation of (a) cubic perovskite with Pm3m symmetry and (b) orthorhombic perovskite with Phnm symmetry. The " $A$ " cation is shown as a sphere.

state for GdAlO<sub>3</sub> has parameters of  $K_{T0} = 191(1)$  GPa and  $K'_0 = 5.8(3)$  compared to  $K_{T0} = 182(1)$  GPa and  $K'_0 = 6.3(3)$  for GdFeO<sub>3</sub>. Analysis of the unit-cell parameter data showed that [100] is least compressible in both compounds and that  $GdFeO<sub>3</sub>$  compresses more isotropically than  $GdAIO<sub>3</sub>$ . While there is no significant change in the pseudo-cubic unit cell parameters with pressure in  $GdFeO<sub>3</sub>$ , they appear to converge in GdAlO<sub>3</sub>. In particular,  $a_c$  and  $b_c$  are predicted to merge by 12 GPa, signifying a possible transition from orthorhombic to tetragonal symmetry [\[6\].](#page-7-0)

Because the compression mechanism in perovskite is largely related to pressure-induced distortions in the  $AO_{12}$  and  $BO_6$  sites that influence the octahedral tilting (e.g. Ref. [\[8\]](#page-7-0)), it is critical that individual bond lengths be determined with increasing pressure. However, as pointed out by O'Keeffe et al. [\[9\]](#page-7-0), the changes in distortion of a perovskite under even 10 GPa of pressure change are expected to be small—they estimated about  $1^{\circ}$  of tilt—and this was for a long time beyond the achievable precision of even single-crystal diffraction methods at high pressures. Recently, several improvements have been made to attempt to reduce the experimental uncertainties and systematic errors in high-pressure determination to the level approaching those obtained from crystals in air. This has enabled the evolution of distortion and tilting of polyhedra in relatively stiff materials including  $YAlO<sub>3</sub>$  [\[10\]](#page-7-0) and

 $CaSnO<sub>3</sub>$  [\[11\]](#page-7-0) to be determined up to pressures of 8 GPa. These studies have also shown that perovskites that exhibit similar distortions from the cubic aristotype at ambient pressure may evolve very differently when subjected to even quite modest pressures.

We report here the structures of  $GdAIO<sub>3</sub>$  and  $GdFeO<sub>3</sub>$ perovskite at high pressure and we address the questions of how the tilting and distortion of the  $AIO<sub>6</sub>$  and  $FeO<sub>6</sub>$ octahedra change with pressure, the atomistic factors that control relative compressibility of the  $GdO_{12}$  and octahedral sites, and how these affect the overall changes in the distortion of the structure.

# 2. Experimental methods

Synthetic  $GdAIO<sub>3</sub>$  and  $GdFeO<sub>3</sub>$  samples were kindly supplied by the Division of Mineralogy, Smithsonian Museum. Single crystals for structural study were selected after X-ray diffraction measurements in air that confirmed that no twinning was obviously present. Experimental details of high-pressure structure determinations are given below.

A GdAlO<sub>3</sub> plate with dimensions of  $178 \times 178 \,\text{\ensuremath{\mu}m}$ was oriented as close as parallel to (110) and polished to  $\sim$ 30  $\mu$ m thickness in order to gain access to the maximum number of reflections along the crystallographic axes. Similarly, a  $(110)$  plate of GdFeO<sub>3</sub> with dimensions  $155 \times 130 \,\mu m$  was polished to  $\sim 32 \,\mu m$ thickness. Each crystal was loaded with (110) parallel to the surface of a  $600 \mu m$  anvil of an ETH diamond anvil cell [\[12\]](#page-7-0) and a 4:1 methanol:ethanol mixture served as the pressure-transmitting medium. A  $200 \mu$ m-thick T301 steel gasket was preindented to a thickness of  $100 \mu m$  for GdAlO<sub>3</sub> and 110 µm GdFeO<sub>3</sub> and holes of  $\phi = 440$ and  $328 \mu m$ , respectively, were drilled in the center of the indented gasket. A ruby sphere was loaded into the cavity to serve as a pressure calibrant [\[13\].](#page-7-0) Intensity data for all accessible reflections in  $GdAIO<sub>3</sub>$  were collected at ambient pressure (in the DAC) and at 1.49(1), 2.47(2), 3.71(3), 5.17(3), 6.49(2), 7.62(2) and 8.52(3) using o scans with the fixed- $\phi$  mode [\[14\]](#page-7-0) from 2° to 40° in  $\theta$  on an Xcalibur diffractometer (Mo $K\alpha$ , 50 kV, 40 mA). Intensity data for  $GdFeO<sub>3</sub>$  were collected at ambient pressure (in the DAC) and at 1.01, 1.99, 3.43, 4.55, 5.94, 6.92, 7.58 and  $8.13$  GPa using the same conditions as GdAlO<sub>3</sub>. Unit cell parameters at these pressures were also measured on a Huber four-circle diffractometer using the 8-position-centering technique  $[14]$  and these were used in the structure refinements. The pressures calculated from the equation of state [\[6\]](#page-7-0) are in good agreement with those measured with the ruby fluoresence method. We determined the offset of the crystal from the rotation axis of the goniometor by measuring between 20 and 40 strong low-angle reflections and calculating the crystal offsets from the reflection

positions with the integration program of Angel WinIntegrStp 3.4 software [\[15\]](#page-7-0). We found that it was critical to eliminate these offsets by adjusting the DAC on the goniometer before data collection. Peak fitting and integration of intensities were carried out by using the WinIntegrStp 3.4 software. Other corrections such as absorption effects of DAC, including correction of beryllium plates, diamond anvils, shadowing by the gasket and the sample itself, were made by using ABSORB 6.0 [\[16\]](#page-7-0). After the crystallographically equivalent reflections were averaged, the remaining independent reflections with  $(F>4\sigma(F))$  were used to refine structures with RFINE99, a development version RFINE4 [\[17\].](#page-7-0) Details of the refinements are given in Tables 1 and 2, the refined positions of atoms and displacement parameters are given in [Tables 3](#page-3-0) [and 4.](#page-3-0)

3. Results and discussion

[Fig. 2a](#page-5-0) shows the variation of the Al–O bond lengths in GdAlO<sub>3</sub>,  $R_{ij}(P)$ , with increasing pressure. Slopes of the bond lengths  $(dR_{ii}/dP)$  were obtained from a linear least-squares fitting and the linear compressibilities of the individual bonds were calculated using the relationship,  $\beta R_{ij} = -1/R_{ij(0)} \times dR_{ij}/dP$  where  $R_{ij(0)}$  is the value of  $R_{ii}$  at ambient pressure. The Al–O21 and Al–O22 bond lengths have similar compressibilities of 0.00172(15) and  $0.00183(13)$  GPa<sup>-1</sup>, whereas the shorter bond Al-O1 is slightly less compressible,  $0.00162(7) GPa^{-1}$ . As a consequence, the degree of distortion due to compression of bond lengths within the  $AIO<sub>6</sub>$  octahedra shows a slight decrease with pressure, which is consistent with the calculated the bond-length distortion  $\Delta_i=$  $1/n \times \sum \{ (R_{ij} - R_{av})/R_{av} \}^2 \times 10^3$  ( $R_{av}$  is the average bond





<sup>a</sup>Number of reflections with  $I > 2I_0/\sigma(I_0)$ .

<sup>b</sup>Number of independent reflections with  $F > 4\sigma(F)$ .

 $\epsilon$ Internal residual on F (number of averaged reflections).

Estimated standard deviation of unit weight observation.

eWeight= $(\sigma_i^2(F_i) + p^2 F_i^2)^{-2}$ .

Weighted  $R_w = \left[\sum_{k=1}^{N} w(|F_0| - |F_c|)^2 / \sum_{k=1}^{N} |F_0|^2 \right]^{1/2}$ .

Unweighted  $R_{\text{uw}} = \sum ||F_{\text{o}}|-|F_{\text{c}}||/\sum |F_{\text{o}}|$ .





<sup>a</sup>Number of reflections with  $I > 2I_0/\sigma(I_0)$ .

 ${}^{b}R_{int}$  Number of independent reflections with  $F > 4\sigma(F)$ .

 $\epsilon$ Internal residual on F (number of averaged reflections).

Estimated standard deviation of unit weight observation.

e Number of averaged reflections.

Funded  $R_w = \left[\sum w(|F_0| - |F_0|^2)^2 / \sum |F_0|^2\right]^2$ .<br>
El Inveighted  $R_v = \sum |F_1| + |F_2| / \sum |F_1|^2$ .

Unweighted  $R_{\text{uw}} = \sum ||F_{\text{o}}|-|F_{\text{c}}||/\sum |F_{\text{o}}|$ .

<span id="page-3-0"></span>Table 3

Unit cell parameters, refined positional parameters and anistropic temperature factors  $(\beta_{ij})$  and equivalent isotopic temperature factors  $(B_{eq})$  of GdAlO3 perovskite at high pressure

$P$ (GPa)	0.0001	1.49(1)	2.47(2)	3.71(3)	5.17(3)	6.49(2)	7.62(2)	8.52(3)
$a(\AA)$	5.2537(1)	5.2421(2)	5.2359(2)	5.2278(2)	5.2186(3)	5.2111(3)	5.2046(2)	5.1999(2)
b(A)	5.3039(1)	5.2858(1)	5.2748(1)	5.2615(2)	5.2461(2)	5.2330(2)	5.2223(2)	5.2136(1)
$c(\check{A})$	7.4435(2)	7.4292(1)	7.4180(1)	7.4041(1)	7.3883(2)	7.3744(2)	7.3625(1)	7.3533(1)
$V(\AA^3)$	207.414(9)	205.852(7)	204.872(7)	203.655(8)	202.29(1)	201.10(1)	200.107(7)	199.347(7)
$Gd^a$								
$\boldsymbol{x}$	$-0.00822(6)$	$-0.00801(7)$	$-0.00780(7)$	$-0.00752(7)$	$-0.00724(8)$	$-0.00714(8)$	$-0.00694(8)$	$-0.00677(8)$
у	0.03770(8)	0.03658(8)	0.03585(8)	0.03498(10)	0.0339(1)	0.0330(1)	0.0323(1)	0.0316(1)
$B_{\text{eq}}$	0.43(1)	0.43(1)	0.46(1)	0.48(1)	0.45(2)	0.45(2)	0.45(2)	0.49(2)
$\beta$ 11	0.0043(3)	0.0039(3)	0.0040(3)	0.0044(3)	0.0033(3)	0.0036(3)	0.0044(3)	0.0050(3)
$\beta$ 22	0.0038(2)	0.0042(3)	0.0046(3)	0.0048(3)	0.0052(3)	0.0051(3)	0.0043(3)	0.0044(4)
$\beta$ 33	0.00183(6)	0.00173(7)	0.00192(7)	0.00189(7)	0.00190(8)	0.00199(8)	0.00190(8)	0.00218(9)
$\beta$ 12	$-0.0006(7)$	$-0.00041(8)$	$-0.00038(8)$	$-0.00031(8)$	$-0.00035(9)$	$-0.00024(7)$	$-0.00025(8)$	$-0.00034(8)$
$Al^b$								
$B_{eq}$	0.36(4)	0.40(3)	0.45(4)	0.38(5)	0.44(5)	0.36(5)	0.40(5)	0.45(5)
$\beta$ 11	0.003(2)	0.0036(8)	0.004(2)	0.004()	0.003(2)	0.002(2)	0.003(2)	0.003(2)
$\beta$ 22	0.003(1)	0.0021(3)	0.004(1)	0.004()	0.005(2)	0.004(2)	0.003(2)	0.005(2)
$\beta$ 33	0.0016(3)	$-0.0002(4)$	0.0020(3)	0.0015(4)	0.0020(4)	0.0017(4)	0.0024(4)	0.0021(5)
$\beta$ 12	0.0001(3)	$-0.0003(4)$	$-0.0007(4)$	$-0.0000(4)$	0.0005(5)	0.0005(4)	0.0002(4)	$-0.0003(4)$
$\beta$ 13	$-0.0003(4)$	0.0006(4)	$-0.0000(4)$	0.0001(4)	0.0001(4)	$-0.0002(4)$	0.0002(4)	$-0.0002(4)$
$\beta$ 23	0.0000(4)	0.0036(8)	$-0.0000(4)$	$-0.0003(4)$	0.0007(4)	0.0004(4)	0.0003(4)	0.0000(5)
$O1^c$								
$\boldsymbol{x}$	0.074(1)	0.073(1)	0.071(1)	0.071(1)	0.071(1)	0.072(2)	0.071(1)	0.070(2)
y	0.486(1)	0.486(1)	0.486(1)	0.487(1)	0.491(1)	0.486(1)	0.489(1)	0.489(1)
$B_{eq}$	0.43(7)	0.41(8)	0.59(9)	0.47(9)	0.6(1)	0.9(1)	0.7(1)	0.6(1)
$\beta$ 11	0.003(2)	0.002(4)	0.005(4)	0.007(4)	0.005(5)	0.012(5)	0.006(4)	0.001(4)
$\beta$ 22	$0.0052(*)$	0.005(4)	0.007(4)	0.0027(4)	0.009(4)	0.009(4)	0.008(5)	0.008(4)
$\beta$ 33	0.0015(7)	0.0021(7)	0.0023(6)	0.0013(7)	0.001(7)	0.0014(7)	0.0028(8)	0.0036(9)
$\beta$ 12	0.0008(9)	$-0.001(1)$	$-0.001(1)$	$-0.001(1)$	0.001(1)	0.000(1)	$-0.001(1)$	$-0.001(1)$
O2								
$\boldsymbol{\chi}$	0.7149(6)	0.7143(7)	0.7152(7)	0.7153(7)	0.7148(7)	0.7148(7)	0.7157(6)	0.7168(7)
у	0.2847(7)	0.2848(7)	0.2850(7)	0.2852(7)	0.2860(6)	0.2854(6)	0.2841(6)	0.2833(6)
$\overline{z}$	0.0397(4)	0.0395(5)	0.0388(5)	0.0389(5)	0.0383(5)	0.0374(6)	0.0387(6)	0.0379(5)
$B_{\rm eq}$	0.41(5)	0.48(6)	0.53(6)	0.50(6)	0.46(7)	0.55(7)	0.51(6)	0.45(7)
$\beta$ 11	0.004(2)	0.006(3)	0.008(3)	0.005(3)	0.002(3)	0.007(3)	0.007(2)	0.004(2)
$\beta$ 22	0.003(2)	0.003(2)	0.003(3)	0.005(3)	0.006(3)	0.004(3)	0.003(2)	0.003(1)
$\beta$ 33	0.0019(4)	0.0019(4)	0.0021(4)	0.0017(4)	0.0021(4)	0.0022(4)	0.0021(4)	0.0018(3)
$\beta$ 12	$-0.0010(8)$	$-0.0016(8)$	0.0011(8)	$-0.0008(9)$	$-0.0018(9)$	0.002(1)	$-0.0006(9)$	$-0.0014(9)$
$\beta$ 13	0.0010(6)	0.0006(6)	$-0.0000(6)$	0.0007(7)	$-0.0002(7)$	0.0012(8)	0.0003(8)	0.0021(9)
$\beta$ 23	$-0.0007(6)$	$-0.0014(6)$	$-0.0006(6)$	$-0.0004(7)$	$-0.0015(8)$	$-0.0009(8)$	0.0000(8)	$-0.0021(8)$

(), The value from the measurement in air was used to avoid non-positive thermal parameters for O1.

<sup>4</sup>Y:  $z = 0.25$ ;  $\beta 13 = \beta 23 = 0$ .

<sup>b</sup>Al:  $x = 0.0, y = 0.5, z = 0.5$ .

 $^{\circ}$ O1:  $z = 0.25$ ;  $\beta$ 13 =  $\beta$ 23 = 0.

length,  $R_{ij}$  is an individual bond length in the *i* site and *n* is the number of bonds) [\[5\].](#page-7-0) For  $AIO<sub>6</sub>$  in  $GdAIO<sub>3</sub>$ at ambient pressure,  $\Delta_{\text{Al}}=0.0036$  and decreases to  $\Delta_{\text{Al}}$ =0.005 at 8.52 GPa. The bond-angle variance ( $\sigma$ ) parameter [\[18\]](#page-7-0) that is equal to zero for a regular octahedron showed a small increase from 0.59 at ambient pressure to 0.62 at 8.52 GPa.

[Fig. 2b](#page-5-0) displays the variation of the Fe–O bond lengths in  $GdFeO<sub>3</sub>$  with increasing pressure. Three Fe–O bond lengths have similar linear compressibilities within the resolution of experiment with  $0.00185(6)$  GPa<sup>-1</sup> for Fe–O1,  $0.00177(17)$  GPa<sup>-1</sup> for Fe–O21, and

 $0.00164(17)$  GPa<sup>-1</sup> for Fe-O22. The FeO<sub>6</sub> octahedra in GdFeO<sub>3</sub> are considerably more distorted than the  $AIO<sub>6</sub>$ in  $GdAIO<sub>3</sub>$  at room pressure, as shown by bond length distortion parameter,  $\Delta_{\text{Fe}}=0.114$ , and bond angle variance parameter,  $\sigma = 1.48$ . The value of  $\Delta_{\text{Fe}}$  at 8.13 GPa is 0.115, indicating no significant change in the bond-length distortion over the pressure range studied, but the  $\sigma$  decreases to 0.92 at 8.13 GPa.

[Fig. 3](#page-5-0) shows the variation of the Gd–O bond lengths in  $GdAIO<sub>3</sub>$  with increasing pressure. The compression of the  $GdO_{12}$  dodecahedral site is more anisotropic than the  $AIO_6$  octahedron. The four longer Gd–O distances

Table 4

Unit cell parameters, refined positional parameters and anistropic temperature factors  $(\beta_{ij})$  and equivalent isotopic temperature factors  $(B_{eq})$  of GdFeO3 peroskite at high pressure

P(GPa)	0.0001	1.01(3)	1.99(7)	3.43(3)	4.55(8)	5.94(4)	6.92(6)	7.58(4)	8.13(9)
a(A)	5.3515(1)	5.3421(3)	5.3326(3)	5.3203(3)	5.3102(4)	5.2986(3)	5.2906(4)	5.2855(3)	5.2812(3)
b(A)	5.6127(2)	5.6020(1)	5.5923(1)	5.5790(1)	5.5676(1)	5.5555(1)	5.5472(2)	5.5418(1)	5.5371(1)
c(A)	7.6713(2)	7.6570(2)	7.6434(1)	7.6259(1)	7.6104(2)	7.5946(1)	7.5838(2)	7.5769(1)	7.5705(1)
$V(\AA^3)$	230.42(2)	229.15(2)	227.94(1)	226.35(1)	225.00(2)	223.55(1)	222.57(2)	221.94(1)	221.38(1)
$\mathbf{Gd}^{\mathrm{a}}$									
$\boldsymbol{\chi}$	$-0.01537(5)$	$-0.01545(6)$	$-0.01549(6)$	$-0.01540(6)$	$-0.01529(6)$	$-0.01534(6)$	$-0.01531(6)$	$-0.01532(6)$	$-0.01538(6)$
$\mathcal{Y}$	0.06255(5)	0.06255(5)	0.06240(5)	0.06232(4)	0.06228(6)	0.06229(5)	0.06221(7)	0.06204(6)	0.06221(5)
$B_{eq}$	0.520(9)	0.52(1)	0.51(1)	0.50(1)	0.49(1)	0.48(1)	0.48(1)	0.48(1)	0.48(1)
$\beta$ 11	0.0043(2)	0.0048(2)	0.0052(2)	0.0049(2)	0.0050(2)	0.0044(2)	0.0045(2)	0.0046(2)	0.0046(2)
$\beta$ 22	0.0043(1)	0.0041(1)	0.0039(1)	0.0039(1)	0.0039(1)	0.0041(1)	0.0040(1)	0.0039(1)	0.0038(1)
$\beta$ 33	0.00219(4)	0.00207(5)	0.00197(5)	0.00195(5)	0.00187(6)	0.00189(5)	0.00195(7)	0.00196(6)	0.00193(5)
$\beta$ 12	$-0.0005(5)$	$-0.00048(5)$	$-0.00040(5)$	$-0.00049(5)$	$-0.00052(6)$	$-0.00052(6)$	$-0.00045(6)$	$-0.00057(6)$	$-0.00040(6)$
Feb Beq $\beta$ 11	0.44(1) 0.0040(4)	0.45(2) 0.0041(5)	0.43(2) 0.0044(5)	0.425(16) 0.0043(5)	0.41(2) 0.0037(6)	0.38(2) 0.0031(5)	0.41(2) 0.0041(6)	0.39(2) 0.0031(6)	0.38(2) 0.0041(6)
$\beta$ 22	0.0040(3)	0.0039(3)	0.0036(3)	0.0035(3)	0.0035(3)	0.0037(3)	0.0034(4)	0.0037(3)	0.0029(3)
$\beta$ 33	0.0015(1)	0.0016(1)	0.0015(1)	0.0015(1)	0.0017(1)	0.0015(1)	0.0015(1)	0.0015(1)	0.0014(1)
$\beta$ 12	0.0001(1)	$-0.0000(1)$	$-0.0003(1)$	0.0001(1)	0.0002(1)	$-0.0001(1)$	$-0.0000(2)$	0.0000(2)	0.0001(2)
$\beta$ 13	0.00007(9)	$-0.0001(1)$	$-0.0000(1)$	$-0.0002(1)$	$-0.0001(1)$	0.0000(1)	0.0002(1)	0.0001(1)	$-0.0001(1)$
$\beta$ 23	0.0002(1)	0.0003(1)	0.0001(1)	0.0001(1)	0.0000(2)	$-0.0001(1)$	0.0002(2)	0.0000(1)	0.0002(1)
$O1^c$									
$\boldsymbol{\chi}$	0.0996(8)	0.1010(9)	0.1006(9)	0.100(1)	0.099(1)	0.1001(9)	0.098(1)	0.100(1)	0.098(1)
$\mathcal{Y}$	0.4677(6)	0.4692(7)	0.4700(6)	0.4699(6)	0.4697(8)	0.4704(8)	0.4713(8)	0.4711(7)	0.4703(8)
$B_{eq}$	0.55(6)	0.68(7)	0.49(4)	0.53(6)	0.51(7)	0.47(7)	0.46(7)	0.54(7)	0.62(7)
$\beta$ 11	0.004(2)	0.007(2)	0.004(2)	0.004(2)	0.005(2)	0.002(2)	0.004(3)	0.005(3)	0.005(2)
$\beta$ 22	0.006(1)	0.006(1)	0.006(1)	0.006(1)	0.005(1)	0.006(1)	0.005(1)	0.007(1)	0.009(1)
$\beta$ 33	0.0020(4)	0.0017(4)	0.0015(4)	0.0016(5)	0.0013(5)	0.0020(5)	0.0012(5)	0.0012(5)	0.0012(5)
$\beta$ 12	$-0.0001(8)$	0.0003(9)	$-0.0002(9)$	$-0.0008(9)$	0.001(1)	0.000(1)	$-0.000(1)$	$-0.001(1)$	$-0.002(1)$
O <sub>2</sub>									
$\boldsymbol{\chi}$	0.6956(6)	0.6945(6)	0.6954(6)	0.6949(6)	0.6937(6)	0.6946(6)	0.6938(6)	0.6948(6)	0.6940(6)
у	0.3012(4)	0.3009(4)	0.3009(4)	0.3010(4)	0.3011(5)	0.3016(5)	0.3018(6)	0.3014(5)	0.3016(5)
$\boldsymbol{z}$	0.0528(3)	0.0526(3)	0.0526(3)	0.0523(3)	0.0525(4)	0.0525(3)	0.0521(4)	0.0518(3)	0.0512(3)
$B_{eq}$	0.60(4)	0.52(4)	0.67(5)	0.57(4)	0.56(6)	0.56(5)	0.50(5)	0.58(6)	0.39(5)
$\beta$ 11	0.005(1)	0.002(1)	0.008(2)	0.007(1)	0.005(2)	0.006(2)	0.004(2)	0.006(2)	0.004(2)
$\beta$ 22	0.0051(7)	0.0061(8)	0.0046(8)	0.0035(8)	0.005(1)	0.0045(8)	0.005(1)	0.004(1)	0.0021(8)
$\beta$ 33	0.0027(3)	0.0023(3)	0.0022(3)	0.0021(3)	0.0025(4)	0.0022(3)	0.0022(4)	0.0026(4)	0.0021(3)
$\beta$ 12	$-0.0000(6)$	$-0.0004(6)$	$-0.0005(6)$	$-0.0004(5)$	$-0.0002(6)$	$-0.0007(6)$	$-0.0001(7)$	0.0003(6)	0.0001(6)
$\beta$ 13	0.0011(4)	0.0007(4)	0.0002(4)	$-0.0001(4)$	0.0006(5)	0.0004(4)	0.0000(5)	0.0003(5)	0.0002(4)
$\beta$ 23	$-0.0012(3)$	$-0.0009(4)$	$-0.0005(3)$	$-0.0004(4)$	$-0.0006(5)$	$-0.0009(4)$	$-0.0014(5)$	$-0.0012(4)$	$-0.0012(4)$

 ${}^{a}Gd$ :  $z = 0.25$ ;  $\beta 13 = \beta 23 = 0$ .<br> ${}^{b}Eq$ :  $x = 0.0$ ,  $y = 0.5$ ,  $z = 0.5$ .

<sup>b</sup>Fe:  $x = 0.0, y = 0.5, z = 0.5$ .

 $^{\circ}$ O1:  $z = 0.25$ ;  $\beta$ 13 =  $\beta$ 23 = 0.

are more compressible (with an average  $\beta_{\text{GdO}}$  of  $0.0028(2)$  GPa<sup>-1</sup>) than the eight shorter Gd-O bond distances (average  $\beta_{\text{GdO}}$  of 0.0010(3)  $\text{GPa}^{-1}$ ) and thus the distortion of  $GdO_{12}$  decreases with increasing pressure. The  $\Delta_{\text{Gd}}$  decreases from 13.33 to 11.63 in the  $GdO_{12}$  polyhedron between room pressure and 8.52 GPa. The bond-angle variance  $(\sigma)$  parameter shows a small increase from 33.78 at room pressure to 34.22 at 8.52 GPa.

The  $GdO_{12}$  site in  $GdFeO_3$  also demonstrates anisotropic compression, as shown in [Fig. 4](#page-6-0). In general, the four longer Gd–O distances are more compressible than the eight shorter Gd–O bond distances with the mean linear compressibility of the four longer Gd–O distances equal to 0.0019(1) GPa<sup>-1</sup>), compared to  $\beta_{\text{GdO8}}$ of  $0.0015(1)$  GPa<sup>-1</sup>. The GdO<sub>12</sub> site becomes less distorted with  $\Delta$  for GdO<sub>12</sub> site decreasing from 28.47 at room pressure to 27.78 at 8.13 GPa. The bond-angle variance  $(\sigma)$  parameter does not show any obvious variation from 32.63 at room pressure to 32.64 at 8.13 GPa.

In order to compare the relative compression of the  $GdO_{12}$  site with the  $AIO_6$  site, the polyhedral bulk moduli,  $K_{\rm P}$ , and compressibilities ( $\beta = 1/K_{\rm P}$ ) of AlO<sub>6</sub>

Fe–O21 bond (circle symbols) and Fe–O22 (triangle symbols) as a function of pressure.

perovskite, Al–O1 bond (square symbols): Al–O21 bond (circle symbols) and Al–O22 (triangle symbols) and (b) the Fe–O bond distances in GdFeO<sub>3</sub> perovskite: Fe-O1 bond (square symbols),

and  $GdO_{12}$  were obtained by fitting the polyhedral volumes with the Birch–Murnaghan finite-strain formulism:

$$
P = \frac{3}{2} K_{\rm P} \left[ \left( \frac{V_{0,\rm poly}}{V} \right)^{7/3} - \left( \frac{V_{0,\rm poly}}{V} \right)^{5/3} \right] \times \left\{ 1 + \frac{3}{4} (K_{\rm P}' - 4) \left[ \left( \frac{V_{0,\rm poly}}{V} \right)^{2/3} - 1 \right] \right\}
$$
 (1)

using the EOSFit v5.2 program [\[14\]](#page-7-0), where  $V_{0,\text{poly}}$  is a polyhedral volume at room pressure,  $K_{P}$  is the polyhedral bulk modulus, and  $K'_{\rm P}$  is the first derivative with respect to pressure (assumed to be equal to 4). The resulting  $V_{0,\text{poly}}$  and  $K_{\text{P0}}$  are 9.222(1)  $\AA$ <sup>3</sup> and 179(8) GPa for AlO<sub>6</sub> and 42.620(26)  $A^3$  and 204(5) GPa for  $GdO_{12}$ . The volume compressibility of the  $GdO_{12}$  site  $(\beta_{\text{GdO12}} = 1/K_{\text{P}})$ ,  $\beta_{\text{GdO12}} = 0.0049(1) \text{ GPa}^{-1}$ , is therefore  $\sim$ 12% less than that of the AlO<sub>6</sub> octahedron  $(\beta_{AIO6} = 0.0056(3) \text{GPa}^{-1})$ . Polyhedral bulk moduli of FeO<sub>6</sub> and GdO<sub>12</sub> in GdFeO<sub>3</sub> obtained from Eq. (1),

Fig. 3. (a) The variation of the eight shortest Gd–O bond lengths in GdAlO<sub>3</sub> perovskite as a function of pressure: Gd–O11 bond (hollow square symbols), Gd–O12 bond (hollow circle symbols), Gd–O21 bond (solid triangle symbols), Gd–O22 bond (solid square symbols), and Gd–O23 bond (solid diamond symbols). (b) The variation of the four longer Gd–O distances as a function of pressure: Gd–O12 bond (hollow square symbols), Gd–O14 (hollow circle symbols) and Gd–O24 (solid triangle symbols).

assuming  $K'_{\rm P} = 4$ , are  $V_{0, \text{poly}} = 10.891(14) \text{ Å}^3$  and  $K_{\rm P0}$  of 188(10) GPa for FeO<sub>6</sub> and  $V_{0,\text{poly}} = 56.595(12) \text{ Å}^3$  and 187.7(18) GPa for  $GdO_{12}$ . The volume compressibility of the GdO<sub>12</sub> site  $(\beta_{\text{GdO12}} = 0.0053(1) \text{ GPa}^{-1})$  is therefore the same within the experimental resolution of the FeO<sub>6</sub> octahedron  $(\beta_{\text{FeO6}} = 0.0053(3) \text{ GPa}^{-1})$ .

As a consequence of the relative compression of the AlO<sub>6</sub> vs. GdO<sub>12</sub> sites, the interoctahedral Al–O1–Al  $(\alpha_1)$ and Al–O2–Al  $(\alpha_2)$  angles both increase with increasing pressure in  $GdAIO<sub>3</sub>$  [\(Fig. 5a](#page-6-0)). Other angle parameters have been introduced [\[19,20\]](#page-7-0) to describe the tilting such as the tilt of the octahedra about the pseduo-cubic  $\langle 110 \rangle_p$  axis,  $\theta$ , and the tilt of the of the octahedra about the pseduo-cubic  $\langle 001 \rangle_p$  axis,  $\phi$  [\(Fig. 5b\)](#page-6-0), that are equivalent to the rotation angle,  $\Phi$ , introduced by O'Keeffe et al. [\[9\]](#page-7-0). From [Fig. 5b](#page-6-0), we see that both  $\phi$  and  $\theta$  decrease slightly within the resolution of the measurement with pressure, which results in, for example, displacement of O1 atoms along  $\langle 100 \rangle$ . All tilt parameters indicate that the structure is becoming less distorted with increasing pressure.

<span id="page-5-0"></span>



<span id="page-6-0"></span>

Fig. 4. (a) The variation of the eight shortest Gd–O bond lengths in GdFeO<sub>3</sub> perovskite as a function of pressure, Gd–O11 bond (hollow square symbols), Gd–O12 bond (hollow circle symbols), Gd–O21 bond (solid triangle symbols), Gd–O22 bond (solid square symbols), and Gd–O23 bond (solid diamond symbols). (b) The variation of the four longer Gd–O distances as a function of pressure, Gd–O12 bond (hollow square symbols), Gd–O14 (hollow circle symbols) and Gd–O24 (solid triangle symbols).

[Fig. 6a](#page-7-0) shows pressure evolution of the interoctahedral angles  $\angle$  Fe–O1 (4c)–Fe ( $\alpha_1$ ) and  $\angle$  Fe–O2–Fe ( $\alpha_2$ ) in  $GdFeO<sub>3</sub>$  as a function of pressure. Both angles show a slight increase with pressure. The  $\alpha_2$  angle increases slightly with increasing pressure whereas  $\alpha_1$  displays a greater increase with pressure, indicating the octahedral  $FeO<sub>6</sub>$  tilting is mainly controlled by variation of  $\angle$  Fe–O1–Fe ( $\alpha_1$ ). As shown in [Fig. 6b,](#page-7-0) the corresponding tilt angle  $\theta$  decreases, whereas  $\phi$  slightly increases with pressure within the resolution of the experiment. Therefore, the octahedral tilting is mainly controlled by the angle,  $\theta$ , which is closely related to the compression of bond Fe–O1 and results in displacement of O1 atoms along  $\langle 100 \rangle$ .

From the results, it is clear that  $GdAIO<sub>3</sub>$  displays a greater change in the degree of distortion than  $GdFeO<sub>3</sub>$ , becoming less distorted than  $GdFeO<sub>3</sub>$  over a similar pressure range. This is consistent with Zhao et al.'s [\[22\]](#page-7-0) model for prediction of high-pressure behavior in Phnm perovskites. Ross et al. [\[6\]](#page-7-0) suggested that a phase



Fig. 5. Pressure evolution of (a) the octahedral tilt angles, Al–O1–Al (solid square symbols) and Al–O2–Al (solid circle symbols) of GdAlO<sub>3</sub> perovskite at high pressure, where the lines represent angles calculated using:  $\alpha_i = 2a \sin[\exp(\Delta \beta_i P) \sin(\alpha_{0i}/2)]$  and (b) corresponding titling angles  $\theta$  and  $\varphi$ .

transition might occur in  $GdAIO_3$  around 12 GPa on the basis of the variation of the unit cell parameters with pressure, and the structural studies of this study verify that the structure is becoming less distorted with pressure and could be approaching a phase transition. Further work is in progress to determine whether a phase transition occurs in  $GdAIO<sub>3</sub>$  perovskite at pressures in excess of 10 GPa.

## 4. Conclusion

The evolution of the atomic-scale structure of  $GdAIO<sub>3</sub>$  and  $GdFeO<sub>3</sub>$  involves compression of both the  $GdO_{12}$  and the  $AlO_6/FeO_6$  sites. The compression of the  $GdO_{12}$  site is anisotropic with the four longest  $Gd-O$ distances more compressible than the eight shorter Gd–O bond lengths, and distortion within  $GdO_{12}$ decreases with pressure. The  $GdO_{12}$  site is less compressible than the  $AIO<sub>6</sub>$  site in GdAlO<sub>3</sub>, resulting in an increase of both the Al–O1–Al and Al–O2–Al angles with increasing pressure. Thus  $GdAIO_3$  perovskite becomes less distorted with increasing pressure. The  $GdO_{12}$  site displays a similar compressibility as the FeO<sub>6</sub> site in  $GdFeO<sub>3</sub>$ , with little change in Fe–O2–Fe with pressure but an increase of the Fe–O1–Fe tiltingwith

<span id="page-7-0"></span>

Fig. 6. Pressure evolution of (a) the octahedral tilt angles, Fe–O1–Fe (solid square symbols) and Fe–O2–Fe (solid circle symbols) of  $GdFeO<sub>3</sub>$ perovskite at high pressure, where the lines represent angles calculated using:  $\alpha_i = 2a \sin[\exp(\Delta \beta_i P) \sin(\alpha_{0i}/2)]$  [10,11] and (b) corresponding titling angles  $\theta$  and  $\varphi$ .

pressure. Thus  $GdFeO<sub>3</sub>$  perovskite becomes less distorted with increasing pressure, but it is not as pronounced as  $GdAIO<sub>3</sub>$ . The high-pressure behavior of  $GdAIO<sub>3</sub>$  and  $GdFeO<sub>3</sub>$  perovskite is therefore similar to YAlO<sub>3</sub> [10,21] but contrasts greatly with  $CaSnO<sub>3</sub>$ , another  $GdFeO<sub>3</sub>$ -type perovskite. Zhao et al. [11] found that the  $SnO_6$  site is less compressible than the  $CaO_{12}$ site in  $CaSnO<sub>3</sub>$  and that the Sn–O–Sn angles decrease with pressure. The distortion of  $CaSnO<sub>3</sub>$  therefore increases with increasing pressure. The different response of the 3-3 and 2-4 perovskites to pressure can be ascribed to the relative compression of the  $AO_{12}$  and  $BO<sub>6</sub>$  sites and their high-pressure behaviors are consistent the model of Zhao et al. [22].

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